A Resource Allocation Framework for Experiment-Based Validation of Numerical Models

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A Resource Allocation Framework for Experiment-Based Validation of Numerical Models

S. ATAMTURKTUR1, J. HEGENDERFER1, B. WILLIAMS2, M. EGEBERG1, R. A. LEBENSOHN3, and C. UNAL4

1Glenn Department of Civil Engineering, Clemson University, Clemson, South Carolina, USA
2CS-Division, Los Alamos National Laboratory, Los Alamos, New Mexico, USA
3MTS-Division, Los Alamos National Laboratory, Los Alamos, New Mexico, USA
4D-Division, Los Alamos National Laboratory, Los Alamos, New Mexico, USA

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In experiment-based validation, uncertainties and systematic biases in model predictions are reduced by either increasing the amount of experimental evidence available for model calibration—thereby mitigating prediction uncertainty—or increasing the rigor in the definition of physics and/or engineering principles—thereby mitigating prediction bias. Hence, decision makers must regularly choose between either allocating resources for experimentation or further code development. The authors propose a decision-making framework to assist in resource allocation strictly from the perspective of predictive maturity and demonstrate the application of this framework on a nontrivial problem of predicting the plastic deformation of polycrystals.

Keywords: Bayesian inference, model calibration, uncertainty quantification, predictive maturity, viscoplastic self-consistent, material plasticity models

1. Introduction

Numerical models are approximate representations of real-world phenomena and, thus, simulations invariably suffer from a degree of inaccuracy and imprecision that can be attributed to: (i) incomplete modeling of physics and/or engineering principles, (ii) imprecisely known model input parameters, and (iii) numerical uncertainties incurred while solving the mathematical equations [1, 2].

Incomplete modeling of physics and/or engineering principles, the first factor, refers to the physical phenomena that are either completely unforeseen or that are known, but too complex to incorporate in the model. This incompleteness invariably causes systematic bias in predictions [3–7] and often leads to missing input parameters [8]. Imprecise model parameters, the second factor, are identified by the analyst; however, their precise values (or distributions) remain unknown [9, 10]. These imprecise model parameters are typically the main contributors to the uncertainty in predictions. Numerical uncertainties, the third factor, can be treated by code and solution verification activities that ensure the mathematical equations are solved correctly [11–13]. Verification is a prerequisite to experiment-based validation [14] and, thus, the third factor, which involves numerical uncertainties, is excluded from the scope of the present article.

The primary aim of the model developer then becomes one of reducing the parameter uncertainty and systematic bias in model predictions. These objectives can be achieved by allocating resources to either (i) undertake experimentation to increase the number of physical observations used in the model calibration process or (ii) code development to improve the manner in which physics and/or engineering principles are defined. Expanding the experimental campaign by conducting new experiments can reduce parameter uncertainty and produce a more refined estimate of systematic bias, while improvement in the description of physics and/or engineering principles can lead to a reduction in the systematic bias. An improved, more detailed model may, however, lead to a higher number of uncertain input parameters and increase the prediction uncertainty [8]. Thus, reducing prediction uncertainty and prediction bias are often conflicting objectives [15]. Therefore, the relative benefits of these two routes, further experimentation versus further code development, vary depending upon available experimental measurements and the existing predictive capability of the numerical model.

Considering the finite resources, effectively choosing one approach over the other becomes an issue of efficient allocation of available resources. By focusing on the relative benefits of each approach strictly from the perspective of predictive capability, the authors propose a resource allocation framework that aids in the selection between further experimentation and code development. The application of this proposed framework is demonstrated on the viscoplastic self-consistent (VPSC) material code for modeling plastic deformation.
2. Resource Allocation Framework

Figure 1 illustrates the proposed framework, which begins with an initial numerical model and a starting set of physical measurements. This initial model is imprecise due to parameter uncertainties and, thus, must be calibrated against experiments. Such calibration is possible using a variety of back-calculation techniques. In this study, a statistical inference procedure originally proposed by Kennedy and O'Hagan [7] and formulated into a standard model calibration procedure by Higdon et al. [5] is adopted. This calibration procedure conditions the probability distributions of input parameters to the experimental evidence reducing the uncertainties in the input parameters, and thus leading to a reduction in prediction uncertainty. An increased availability of experiments results in a greater reduction in prediction uncertainty. In the framework proposed herein, this reduction is traced by quantifying the information gained using an entropy-based metric.

Aside from prediction uncertainty, the initial numerical model in Figure 1 also has a prediction bias, i.e., fundamental inability to reproduce the reality that cannot be remedied solely by calibrating the input parameters. According to Kennedy and O'Hagan [7], prediction bias can be determined by quantifying the deviations between the experiments and the model predictions obtained with ‘best fitted’ parameter values. Since experiments are only available at discrete settings, empirically training an error model becomes necessary to estimate the systematic bias at the untested settings [5]. As more experiments become available, the prediction bias can be estimated with greater fidelity, ultimately converging to the “true” bias of the model. In the framework proposed herein, the convergence of prediction bias is traced by calculating an averaged bias for the entire domain of applicability through the independently trained empirical error model, henceforth referred to as discrepancy.

At the decision node in the framework, the decision maker must assess the predictive capability of the numerical model by evaluating the stabilization of the discrepancy and the information gain metrics throughout the domain of applicability. For a sound model developed based on well-founded physics or engineering principles, the absence of stabilization of the information gain metric indicates that the prediction uncertainty has not yet been fully mitigated, while the absence of stabilization of the discrepancy indicates that prediction bias has not yet been properly defined. In this case, resources must be allocated for experimentation. Poorly built models fail to exhibit stabilization in discrepancy even after a significant number of experiments are conducted; therefore, at this point in the framework, a measure of how well the current experiments explore the domain of applicability (referred to as coverage) is compared to a maximum coverage limit [16–18]. Exceeding the maximum coverage limit without stabilization indicates that the model is too crude for its purposes and resources must be allocated for code development.

If stabilization is observed, however, then the ability of the available experiments to sufficiently explore the domain is analyzed. Stabilization, when only a small portion of the domain is explored by experiments, cannot ensure that the prediction bias is properly defined throughout the entire domain. Therefore, a minimum coverage limit needs to be reached; otherwise, more experiments must be conducted to explore the domain of applicability.

If stabilization is observed and the minimum coverage threshold is met, then the epistemic component of parameter uncertainty can be expected to be adequately reduced, and the inferred prediction bias can be considered a proper representation of the incompleteness and inexactness of the model. In this case, further experimentation would only marginally improve the inference of the systematic bias, and allocating resources to experiments cannot be justified. The decision maker must then evaluate if the remaining prediction bias is at an acceptably low level for the application of interest.

If the prediction bias is sufficiently low, the model is considered valid for the particular application. If the prediction bias is unacceptably high for the specific purposes of model predictions, the physics and/or engineering principles in the model must be improved. An improved model may have a larger number of uncertain parameters and may require a more extensive experimental campaign to mitigate the increased prediction uncertainty [8]. Therefore, it becomes necessary to check the stabilization of both information gain metric and discrepancy simultaneously. The framework in Figure 1 therefore loops through the aforementioned steps until the formulated discrepancy and information gain metrics converge to the acceptable levels.

3. Metrics for Prediction Uncertainty and Bias

3.1. Prediction Uncertainty: Information Gain

As the information gain is equal to the amount of uncertainty removed, entropy, defined as a measure of uncertainty,
is equivalent to the amount of information [19]. Herein, an
information gain metric based on Shannon entropy is utilized
to quantify the prediction uncertainty as additional experiments
become available to condition the posterior distributions of
input parameters. For a discrete random variable, \( z \), with a
probability mass function, \( p(z) \), the entropy is expressed as:

\[
H(z) = - \sum_{z \in \mathcal{Z}} p(z) \log p(z),
\]

where the logarithm to base 2 is used to measure entropy in
bits and \( z \) represents the calibrated model predictions. As an
increasing number of experiments are used in the calibration
process, the information gain metric, expressed in percentages,
is calculated using the following relationship:

\[
\text{Info - Gain}(i)\% = \left( \frac{H_{\text{ref}} - H_{\text{exp}(i)}}{H_{\text{ref}}} \right) \times 100,
\]

where \( H_{\text{ref}} \) is the entropy calculated for the model predictions
obtained with the initial distributions of the calibration pa-
rameters prior to the availability of experiments. \( H_{\text{exp}(i)} \) is the
entropy of the calibrated model predictions, in which \( i \) is the
number of experiments used in the calibration process.

While the information gain metric is an excellent tool for
quantifying the reduction in prediction uncertainty, it does
not make any assertions about the prediction bias. Therefore,
both the convergence of systematic bias and information gain
must be evaluated while discerning the necessity of additional
experiments or further code development.

### 3.2. Prediction Bias: Model Form Error

A multivariate generalization [5, 6] of a model calibration
approach formulated by Kennedy and O’Hagan [7] is imple-
mented, in which experiments are exploited to infer uncertain
input parameters while simultaneously considering the incom-
pleteness of the model. Here, the experimental observation,
\( y(x) \), is given by:

\[
y(x) = \eta(x, \theta) + \delta(x) + \varepsilon(x),
\]

where \( \eta(x, \theta) \) denotes the model predictions, \( \delta(x) \) represents
the estimated systematic bias between reality and the predic-
tions, and \( \varepsilon(x) \) denotes the experimental error. Here, \( x \)
represents settings at which observations are made (i.e., control
parameters), and \( \theta \) denotes the best values for the calibration
parameters, \( t \).

For many practical problems, the complexity and computa-
tional demands of numerical models limit the number of
possible runs. To obtain predictions at untried settings, an
inexpensive surrogate (also known as emulator) can be trained as
a substitute for the numerical model. Here, a Gaussian process
(GP) emulator is used to represent the numerical model pre-
dictions, \( \eta(x, t) \), which is specified by a mean function, \( \mu(x, t) \),
and a covariance function [5]:

\[
\text{Cov}((x, t), (x', t')) = \frac{1}{\lambda_{\theta}} \prod_{k=1}^{p_{\theta}} \rho_{\theta k}^4 \left( x_k - x'_k \right)^2
\]

where \( \lambda_\theta \) and \( \rho_{\theta k} \) are hyper-parameters for the GP emulator
for prediction bias, which control the marginal precision of \( \delta(x) \) and
the dependence strength in the components of the \( x \) direction,
respectively. The hyper-parameters ensure a smooth and differentiable form for both \( \delta(x) \) and \( \delta(x) \).

In the Bayesian calibration framework, the true but un-
known values of the calibration parameters, \( \theta \), are inferred ex-
ploring the availability of the experimental data, where the ex-
isting knowledge about calibration parameters and the hyper-
parameters of the GP emulators are incorporated through
prior distributions. The posterior distribution conditioned on
experimental data is given by:

\[
\pi(\theta, \mu, \lambda_\eta, \rho_\eta, \lambda_\delta, \rho_\delta | D) \propto L(D|\theta, \cdot) \prod_{i} \pi(\theta) \times \pi(\mu) \times \pi(\lambda_\eta) \times \pi(\rho_\eta) \times \pi(\lambda_\delta) \times \pi(\rho_\delta),
\]

where \( D \) is the joint vector of experimental data and numerical
model outputs, \( L(D|\theta, \cdot) \) is the likelihood function, \( \sum_{i} \) is the
observation covariance matrix, and \( \pi(\cdot) \) is the prior distribu-
tion(s) (for detailed discussion see Higdon et al. [5]).

A Markov chain Monte Carlo (MCMC) algorithm, specifi-
cally Metropolis–Hasting algorithm [20, 21], is used to explore
the posterior distributions for both the calibration param-
eters and the aforementioned hyper-parameters. During the
MCMC random walk, calibration parameter values that gen-
erate predictions with greater agreement with the experimental
data over the domain of applicability are accepted based on
the established maximum likelihood criteria [20, 21].

Upon obtaining the hyper-parameters of the GP model, the
discrepancy \( \delta(x^*) \) can be estimated at untested input settings,
\( x^* \) throughout the entire domain of applicability. This empir-
ically trained discrepancy model can then be aggregated to
obtain an average representation of prediction bias through-
out the domain.
4. Viscoplastic Self-Consistent (VPSC) Material Model

Lebensohn and Tomé [22] developed a VPSC material model for modeling the plastic deformation of polycrystals. A polycrystal is modeled by a set of single crystals (grains) with initial crystallographic orientations that represent the initial texture of the aggregate and evolve during plastic deformation. In turn, each grain is treated as an ellipsoidal inclusion with anisotropic viscoplastic properties, deforming in a homogenous equivalent medium that has the a priori unknown average properties of the aggregate. This leads to a relation between the strain-rate and stress in each individual grain with the global stress and strain-rate of the aggregate through localization equations. The viscoplastic deformation of the crystals occurs by dislocation motion and can be modeled in terms of constitutive relations between the deviatoric stress and strain-rate tensors. Viscoplastic deformation will occur when a slip system activates and dislocations move under an applied stress. The final deformation is obtained in the VPSC formulation through imposing a macroscopic strain-rate during each incremental deformation step. The strain-rate and stress from each previous step is used as the starting values for the next step. Stress–strain curves and texture development constitute the typical output of a VPSC calculation. Recall that the numerical uncertainty must be verified prior to validation (see Lebensohn et al. [23] for verification of the VPSC code against “exact” full-field formulations).

In the present study, two versions of the VPSC code are utilized: the original glide-only (G) version, used for predictions of plasticity of polycrystals; and the climb-and-glide (C&G) version, with improved physics for the prediction of polycrystal response under creep conditions.

4.1. Glide VPSC

In the G version of the VPSC code, a Schmid-type constitutive behavior is used to describe the dislocation motion in the constituent single-crystals [22]. As such, dislocations lie and move within the slip plane and are activated by shear stresses; their motion can only accommodate simple shear deformation on this plane. Glide activity in several slip planes is able to accommodate an arbitrary deformation applied to the crystal. The constitutive equation at the single crystal level is expressed as:

\[ \dot{\varepsilon} = \dot{\gamma}_o \sum_{s=1}^{N_s} m_s \left( \frac{|m^s : \sigma|}{\tau^s_o} \right)^{n_s} \text{sgn} (m^s : \sigma), \]  

(7)

where \( \sigma \) is the stress applied to the crystal and \( \dot{\varepsilon} \) is the strain rate, accommodated by glide; \( m^s \) and \( \tau^s_o \) are the Schmid tensor and the critical resolved shear stress associated with glide in the system(s), respectively. The stress exponent, \( n_s \), represents the inverse of rate sensitivity for the glide activity, and \( \dot{\gamma}_o \) denotes a normalization factor. The single crystal equation for strain rate is summed over all active slip systems, \( N_s \).

4.2. Climb-and-Glide VPSC

Lebensohn et al.’s [24] constitutive model for aggregates of single crystals deforming by climb and glide is an improvement to the original VPSC approach that considers deformations by glide only. At temperatures below 50% of the melting temperature, glide-controlled creep dominates; however, at higher temperatures, local nonequilibrium concentrations of point defects interacting with dislocations allow for dislocations to climb in addition to glide. Dislocation climb becomes very relevant in high-temperature plasticity and irradiation creep. The direction of dislocation motion is determined by the velocity vector composed of two components: the glide velocity (lies in glide plane) and the climb velocity (normal to the glide plane). The glide component depends upon the shear stress component acting on the glide plane while the climb component depends on the full stress tensor. The extension of Eq. (7) to the C&G case is expressed as:

\[ \dot{\varepsilon} = \dot{\gamma}_o \sum_{s=1}^{N_s} m_s \left( \frac{|m^s : \sigma|}{\tau^s_o} \right)^{n_s} \text{sgn} (m^s : \sigma) \]  

\[ + \dot{c}_o \left( \frac{|c^s : \sigma|}{\tau^s_o} \right)^{n_c} \times \text{sgn} (c^s : \sigma), \]  

(8)

\[ \text{Table 1. Stress and texture intensity experimental results for 5182 Al} \]

<table>
<thead>
<tr>
<th>Experiment</th>
<th>Temperature (°C)</th>
<th>Strain rate (s(^{-1}))</th>
<th>Stress (MPa) @ strain = 0.6</th>
<th>Texture intensity (001)</th>
<th>Texture intensity (101)</th>
</tr>
</thead>
<tbody>
<tr>
<td>A</td>
<td>200</td>
<td>10(^{-3})</td>
<td>226.2</td>
<td>1.00–1.41</td>
<td>4.00–6.00</td>
</tr>
<tr>
<td>B</td>
<td>300</td>
<td>10(^{-3})</td>
<td>91.4</td>
<td>0.58–0.71</td>
<td>4.00–6.00</td>
</tr>
<tr>
<td>C</td>
<td>350</td>
<td>10(^{-3})</td>
<td>50.0</td>
<td>2.00–2.83</td>
<td>2.83–4.00</td>
</tr>
<tr>
<td>D</td>
<td>400</td>
<td>10(^{-3})</td>
<td>30.6</td>
<td>NA</td>
<td>NA</td>
</tr>
<tr>
<td>E</td>
<td>500</td>
<td>10(^{-3})</td>
<td>14.9</td>
<td>4.00–6.00</td>
<td>2.00–2.83</td>
</tr>
<tr>
<td>F</td>
<td>550</td>
<td>10(^{-3})</td>
<td>7.0</td>
<td>NA</td>
<td>NA</td>
</tr>
<tr>
<td>G</td>
<td>200</td>
<td>1</td>
<td>280.0</td>
<td>NA</td>
<td>NA</td>
</tr>
<tr>
<td>H</td>
<td>300</td>
<td>1</td>
<td>193.7</td>
<td>NA</td>
<td>NA</td>
</tr>
<tr>
<td>I</td>
<td>400</td>
<td>1</td>
<td>121.3</td>
<td>1.41–2.00</td>
<td>4.00–6.00</td>
</tr>
<tr>
<td>J</td>
<td>500</td>
<td>1</td>
<td>65.5</td>
<td>2.00–2.83</td>
<td>0.00–0.58</td>
</tr>
<tr>
<td>K</td>
<td>550</td>
<td>1</td>
<td>43.0</td>
<td>2.83–4.00</td>
<td>0.58–0.71</td>
</tr>
</tbody>
</table>
where $c^*$ and $\sigma^*_c$ are, respectively, the climb tensor and a critical stress associated with climb in the system(s), and $n_c$ is the stress exponent associated with climb. In Eq. (8), $n_g$ and $\tau^*_g$ (glide stress exponent and critical resolved shear stress associated with glide) are temperature, strain-rate, and microstructure-dependent [24]. Likewise, $n_c$ and $\sigma^*_c$ (climb stress exponent and critical stress associated with climb) are dependent of the same variables. This latter dependency however, is much more complex due to the dynamics and interactions of dislocations and the interactions between point defects and dislocations. Additional versions of the VPSC code, including an improved C&G model [25] and an atomistic scale coupled model, are currently being developed and will be added to future analysis.

5. Application of the VPSC Code to 5182 Aluminum Alloy

Experiments performed on 5182 aluminum samples with an initial (001) (“cube”) texture deformed in compression have been reported in Stout et al. [26, 27]. Stress–strain curves and final textures (in terms of inverse pole figures) are measured for varying levels of the two control parameters: temperature and strain rate (Table 1). The experiments are performed until the specimens reach a true strain of 0.6. Final textures are also available for 7 of the 11 stress–strain curves shown in Figure 2.

The stress corresponding to the maximum measured strain of 0.6 and the intensities of the textures corresponding to the (001) and (101) corners of the inverse pole figure are extracted as low-dimensional data for the calibration of the VPSC model. Although a complete, quantitative description of crystallographic textures requires, in general, a large number of parameters (e.g., weights associated with a partition of a 3D orientation space), the final compression textures of the 5182 Al samples can be characterized by, at most, two components with associated intensities, corresponding to a retained (001) cube component and/or a (101) compression texture. These chosen features offer a low-dimensional yet highly informative metric for model calibration.

5.1. Experimental Observations

The initial results by Stout et al. [26] show typical stress–strain behaviors, in which yielding is followed by strain-hardening at low temperatures. At higher temperatures, however, very little work hardening and lower yield stresses are observed. Additionally, for yield stresses below 50 MPa, negative work hardening occurs with a clear upper/lower yield point; for stresses above 50 MPa, a yield point is not observable, however [26]. Textures at elevated temperatures are likewise inconsistent with standard glide-only deformation textures, while textures at lower temperatures develop a (101) fiber texture, typical of uniaxial compression applied to a FCC polycrystal. Experiments run at 500 and 550°C with a strain rate of $10^{-3}$ s$^{-1}$ display a (001) cube component (generally thought of as a recrystallization texture not a deformation texture) and almost no (101) deformation component. For 400, 500, and 550°C with 1 s$^{-1}$ strain-rate, both (001) and (101) textures are observed [27]. To explain the difference in the combination of textures, Stout et al. [27] assumes a sharp decrease in rate sensitivity as strain rate increases.

5.2. Prior Work

Utilizing the VPSC model to predict texture measurements reported by Stout et al. [27], Lebensohn et al. [24] observes that the (001) cube component at high temperatures and low strain

### Table 2. Control and uncertain model parameter values

<table>
<thead>
<tr>
<th>Parameter</th>
<th>Minimum</th>
<th>Maximum</th>
</tr>
</thead>
<tbody>
<tr>
<td>Control parameters</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Temperature (°C)</td>
<td>180</td>
<td>570</td>
</tr>
<tr>
<td>Strain-rate (s$^{-1}$)</td>
<td>0.0005</td>
<td>1.05</td>
</tr>
<tr>
<td>Uncertain model parameters</td>
<td></td>
<td></td>
</tr>
<tr>
<td>$n_g$</td>
<td>1</td>
<td>5</td>
</tr>
<tr>
<td>$\tau^*_g$ (MPa)</td>
<td>1.2</td>
<td>1343.4</td>
</tr>
<tr>
<td>$n_c$</td>
<td>1</td>
<td>5</td>
</tr>
<tr>
<td>$\sigma^*_c$ (MPa)</td>
<td>1.2</td>
<td>6045.4</td>
</tr>
</tbody>
</table>
rates are due to an increase in climb activity. Seven different VPSC simulations of texture evolution are computed for various glide-only and climb-and-glide scenarios. An analysis of these simulations shows that an increase in rate-sensitivity contributes to prevalence of the (001) cube component. The texture simulated using an equal climb-to-glide activity ratio most accurately predicts the experimental texture for the 400°C and 10^{-3} s^{-1} experimental case, however. As proposed in Lebensohn et al. [24], the final retained (001) cube component is achieved by means of an increase in climb activity, which can be explained by climb mechanism accommodating plastic deformation applied to a single crystal involving a reduced plastic spin (crystal rotation) as compared with glide (see Lebensohn et al. [24] for details). Hereafter, the response features of interest (stress at the maximum measured strain of 0.6, (001) cube texture, and (101) compression texture) will be referred to as maximum stress, texture 001, and texture 101, respectively.

### 6. Calibration Against Experimental Data

The critical stresses and the stress exponents from Eqs. (7) and (8) are uncertain and, thus, will be calibrated against experimental data. As the initial strain hardening is not of interest, the critical stresses are assumed constant. For the G model, the two parameters that need to be calibrated are then (i) the glide stress exponent, \( n_g \), and (ii) the initial critical resolved shear stress for glide, \( \tau_{0g} \). For the C&G model, in addition to the two parameters corresponding to the glide mechanism, parameters that need calibration include two more parameters: (iii) the climb stress exponent, \( n_c \), and (iv) critical stress associated with climb, \( \tau_{0c} \). The plausible upper and lower limits for these uncertain parameters are determined by expert opinion and are listed in Table 2.

#### 6.1. Correlation Function

There is a potential dependency of the stress exponent(s), \( n \), and initial critical shear stress(es), \( \tau \), on the control parameters (temperature and strain rate) making it implausible to search for a single set of input parameter values for stress exponent and critical stress that can yield satisfactory agreement with experiments throughout the entire domain of applicability. Therefore, it becomes necessary to construct a correlation function to investigate and if present represent the dependency of these uncertain input parameters on control parameters.

<table>
<thead>
<tr>
<th>Parameter</th>
<th>Optimized/mean value</th>
<th>Min</th>
<th>Max</th>
</tr>
</thead>
<tbody>
<tr>
<td>( a_g )</td>
<td>2970.2</td>
<td>2227.65</td>
<td>3712.75</td>
</tr>
<tr>
<td>( b_g )</td>
<td>-0.008</td>
<td>-0.0064</td>
<td>-0.0096</td>
</tr>
<tr>
<td>( c_g )</td>
<td>281.7</td>
<td>211.275</td>
<td>352.125</td>
</tr>
<tr>
<td>( d_g )</td>
<td>-0.004</td>
<td>-0.0032</td>
<td>-0.0048</td>
</tr>
<tr>
<td>( n_g )</td>
<td>3.5</td>
<td>2.5</td>
<td>4.5</td>
</tr>
<tr>
<td>( a_c )</td>
<td>24,727</td>
<td>18,545.25</td>
<td>30,908.75</td>
</tr>
<tr>
<td>( b_c )</td>
<td>-0.012</td>
<td>-0.0096</td>
<td>-0.0144</td>
</tr>
<tr>
<td>( c_c )</td>
<td>1595.2</td>
<td>1196.4</td>
<td>1994</td>
</tr>
<tr>
<td>( d_c )</td>
<td>-0.008</td>
<td>-0.0064</td>
<td>-0.0096</td>
</tr>
<tr>
<td>( n_c )</td>
<td>3.5</td>
<td>2.5</td>
<td>4.5</td>
</tr>
</tbody>
</table>
Table 5. Calibration experiments used in each case (see Table 1 for the experimental settings)

<table>
<thead>
<tr>
<th>Case</th>
<th>Experiments</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>A</td>
</tr>
<tr>
<td>2</td>
<td>A, B</td>
</tr>
<tr>
<td>3</td>
<td>A, B, C</td>
</tr>
<tr>
<td>4</td>
<td>A, B, C, E</td>
</tr>
<tr>
<td>5</td>
<td>A, B, C, E, I</td>
</tr>
<tr>
<td>6</td>
<td>A, B, C, E, I, J</td>
</tr>
<tr>
<td>7</td>
<td>A, B, C, E, I, J, K</td>
</tr>
<tr>
<td>8</td>
<td>A, B, C, E, I, J, K, D</td>
</tr>
<tr>
<td>9</td>
<td>A, B, C, E, I, J, K, D, E</td>
</tr>
<tr>
<td>10</td>
<td>A, B, C, E, I, J, K, D, E, G</td>
</tr>
<tr>
<td>11</td>
<td>A, B, C, E, I, J, K, D, E, G, H</td>
</tr>
</tbody>
</table>

Such a function can be constructed by exploiting the available experimental data.

Optimal values for stress exponent, \( n \), and critical stress, \( \tau \), are obtained by minimizing the disagreement between the measured stress and texture values using a nonlinear constrained optimization algorithm, exploiting the experimental measurements available at settings given in Table 1. Sequential quadratic programming (SQP) through the fmincon command in MATLAB is implemented [28, 29] (also see [30] for a complete description of the gradient-based SQP algorithm).

In optimization of the G model, the stress exponents are allowed to vary between 1 and 5, with starting values of 3 and 3.3 for the glide and climb stress exponents, respectively. The lower bound and upper bound of the range for the critical stresses corresponding to the settings of each experimental measurement are determined using Eq. (9) and Eq. (10), respectively.

\[
\begin{align*}
\tau_{\text{min}} &= \frac{0.5 \times \sigma^e}{3 \times \dot{\varepsilon}^{(1/3.5)}}, \\
\tau_{\text{0max}} &= \frac{2.0 \times \sigma^e}{3 \times \dot{\varepsilon}^{(1/3.5)}}.
\end{align*}
\] (9) (10)

where \( \sigma^e \) represents the stress at the maximum measured strain. When the range identified for these parameters is normalized between 0 and 1, the starting value for the critical stress is set to 0.33.

For optimization of the C&G model, the stress exponents are allowed to vary between 1.5 and 4.5 with the same starting values as the G model. The bounds on the critical stress associated with glide and the lower bound of the critical stress associated with climb are determined according to Eqs. (9) and (10). The upper bound of the critical stress associated with climb is determined from Eq. (11):

\[
\sigma_{\text{0max}} = \frac{9.0 \times \sigma^e}{3 \times \dot{\varepsilon}^{(1/3.5)}}.
\] (11)

When the range identified for these parameters is normalized between 0 and 1, the starting value for the critical stress associated with glide is 0.4 and the initial critical stress associated with climb is 0.9.

The function to be minimized is given by the least-squares objective function in Eq. (12):

\[
O = \sqrt{\left( \frac{\sigma^e(n, \tau) - \sigma^e}{\sigma^e} \right)^2 + \left( \frac{\psi(n, \tau)_{(001)} - \psi^e_{(001)}}{\psi^e_{(001)}} \right)^2 + \left( \frac{\psi(n, \tau)_{(101)} - \psi^e_{(101)}}{\psi^e_{(101)}} \right)^2},
\] (12)

where \( \sigma^e \) represents the predicted stress at the maximum measured strain. In Eq. (12), \( \psi^e_{(001)} \) and \( \psi(n, \tau)_{(001)} \) denote the measured and predicted texture 001 intensities, respectively; while \( \psi^e_{(101)} \) and \( \psi(n, \tau)_{(101)} \) represent the measured and predicted texture 101 intensities, respectively. Note that for each experimental setting given in Table 1, the \( \psi^e_{(001)} \) and \( \psi^e_{(101)} \) values are given as ranges due to the experimental uncertainty.

Therefore, the following relationships are imposed on the

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Fig. 5. Posterior distribution for the G model (Left: 1 experiment, Right: 11 experiments) (Inner contour: 90th percentile, Outer contour: 50th percentile).
second and third terms in Eq. (12):

\[
\left( \frac{|\psi^s - \psi^c|}{\psi^c} \right)^2 = 0 \text{ for } (\psi^c)_L \leq \psi^s \leq (\psi^c)_U, \quad (13)
\]

\[
\left( \frac{|\psi^s - (\psi^c)_L|}{(\psi^c)_L} \right)^2 = \left( \frac{|\psi^s - (\psi^c)_U|}{(\psi^c)_U} \right)^2 \text{ for } \psi^s < (\psi^c)_L, \quad (14)
\]

\[
\left( \frac{|\psi^s - (\psi^c)_L|}{(\psi^c)_L} \right)^2 = \left( \frac{|\psi^s - (\psi^c)_U|}{(\psi^c)_U} \right)^2 \text{ for } \psi^s > (\psi^c)_U. \quad (15)
\]

In Eqs. (13)-(15), the subscripts \( U \) and \( L \) denote the upper and lower bounds for texture data from Table 1. Because increased stress leads to an increase in glide at the expense of climb, the constraint \( n_c < n_g \) is imposed for all control parameter settings.

Representative plots of the optimized, deterministic point estimates for stress exponent(s) and critical stress(es) are plotted in Figures 3 and 4 for G and C&G models, respectively. In these figures, the starting values are 3 and 3.3 for the glide and climb stress exponents, respectively. Other starting values between 1.8 and 4.2 for the G model and between 2.1
and 3.9 for the C&G model yield similar results, demonstrating that the optimized stress exponent(s), n, are independent of temperature and strain rate and, thus, must be treated as calibration parameters. Furthermore, the optimization process indicates that bounds for the stress exponents can be restricted further (see the gray band on Figures 3 and 4) from the ranges identified by the expert (recall Table 2). The further constrained ranges for the stress exponents are listed in Tables 3 and 4.

On the other hand, the optimized critical stress, $\tau$, exhibits an exponential relationship with temperature and a linear relationship with strain rate, which can be expressed as:

$$
\tau = a e^{b T} + \frac{(\dot{e} - \dot{e}_1) c e^{d T} - (\dot{e} - \dot{e}_1) a e^{b T}}{\dot{e}_2 - \dot{e}_1},
$$

(16)

where $\dot{e}$ is the strain rate and $T$ is the temperature. The variables $a$ and $c$ are the leading intercept coefficients for the exponential fit for the critical stress at a strain rate of $\dot{e}_1 = 0.001$ and $\dot{e}_2 = 1$, respectively. Likewise, $b$ and $d$ are the decay rate coefficients for the exponential fit. In lieu of critical stresses, the coefficients $a$, $b$, $c$, and $d$ are treated as calibration parameters, for which posterior distributions are inferred. The optimal values for the coefficients computed from Eq. (12) and shown in Tables 3 and 4 for the two models, are treated as starting values during model calibration (discussed in the next section).

The aforementioned analysis yields 5 calibration parameters for the G model and 10 for the C&G model as listed in Tables 3 and 4, along with the ranges, within which these parameters are allowed to vary to encompass available experimental data.

### 6.2. Model Calibration

Latin-hypercube designs with 140 and 240 samples are used to construct the GP emulators for the model predictions of the G model and the C&G model, respectively. For the calibration parameters, a uniform prior distribution is assumed between the ranges given in Tables 3 and 4. Further, 10,000 accepted MCMC iterations are generated to estimate the posterior distribution of the calibration parameters. Exercising the GP emulators, predictions are obtained for 500 linearly spaced samples from the posterior distributions of the calibration parameters and GP hyper-parameters. To compare the calibrated model predictions against experimental data, predictions, $\eta(x; \theta)$, are generated at experimental settings shown in Table 1.
7. Results and Discussion

This section demonstrates the resource allocation framework and provides a discussion on the computed systematic bias and information gain as the two versions of the VPSC model are calibrated using one through 11 available experiments in the sequence shown in Table 5. When all 11 experiments are used, coverage of the domain reaches over 85% according to the coverage metric proposed in Hemez et al. [17]. It is assumed that the minimum coverage threshold is set at 80% for this application; therefore, the 11 experiments satisfactorily explore the domain of applicability. The calibrated VPSC models are executed to predict maximum stress along with texture 001 and 101 intensities at the settings of these 11 experiments. This section ultimately concludes by comparing the tradeoff of bias and uncertainty reduction between the two models through divergence information criterion.

7.1. Initial Numerical Model: G Model

A reduction in uncertainty in the calibration parameters of the G model can be observed by the narrowing of the distributions in Figure 5, which compares the posterior distributions of input parameters obtained using all 11 experiments to those obtained using only one experiment (with the exception of \(d_g\), which can be explained by the relative insensitivity of this parameter). In Figure 6, the stress and texture predictions depict convergent behavior with consistently reduced uncertainty as the number of experiments used in the analysis increases. Also evident in Figure 6 is the constant systematic bias remaining.

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Fig. 9. Systematic bias corresponding to texture 101 for each experiment: (a) G model, (b) C&G model.

Fig. 10. Information gain from experiments used for calibration of G model: (a) Experiment E, (b) Experiment K.
after the convergence of predictions. The remaining systematic bias is measured to the nearest dashed horizontal line, which reflects the experimental uncertainty.

Figures 7a–9a depict the prediction bias for the maximum stress, texture 001, and texture 101 predictions at each experimental setting, as a function of the number of experiments used in the calibration process. For the maximum stress output shown in Figure 7a, little convergent behavior is evident; while for the texture outputs, upon addition of the fourth experiment, the inferred prediction bias of the textures converges for all prediction settings, as shown in Figures 8a and 9a.

The information gain is also computed for the model predictions at all experimental settings (recall Table 5). For brevity, however, Figure 10 shows the information gain plots for the maximum stress, texture 001, and texture 101 predictions at settings E and K, respectively. The remaining experimental settings show similar trends. For stress

Fig. 11. Model prediction bias and uncertainty for the C&G model (Left: Experiment K, Right: Experiment E).
predictions, the information gain monotonically increases and ultimately converges. Slight fluctuations are expected given the stochastic nature of the inference approach. For the texture predictions, a convergent behavior is not observed, indicating the G model’s inability to reproduce texture experiments at certain regions in the domain of applicability. Furthermore, Figures 8 and 9 depict a prediction bias well in excess of 100% for the texture predictions demonstrating that uncertainty reduction alone is insufficient to warrant accurate predictions. In fact, Figure 9 reveals a systematic bias for texture 101 over 400% for setting K, and as high as 1000% for setting J.

Referring back to Figure 1, the G model’s systematic bias is unacceptably high. The resource allocation framework then suggests that the rigor in which the physics principles are modeled must be improved, which leads to the C&G model.

7.2. Improved Numerical Model: C&G Model

In Figure 11, it can be observed that the uncertainties in stress and texture predictions of the C&G model reduce and the prediction bias converges to a constant value as the number of experiments available for model calibration increases. Comparing Figures 6 and 11, the more sophisticated, C&G model has significantly less prediction bias compared to the G model. However, the C&G model requires an increased number of experiments to reach convergence. For the C&G model, from Figures 8 and 9, seven experiments are necessary for the prediction bias of the texture to converge; while convergence is achieved with only four experiments for the G model. An increased need for experiments to mitigate parameter uncertainty is expected for the C&G model, which has twice as many calibration parameters as the G model (see Tables 3 and 4).

Figure 12 shows the convergence of information gain after a sufficient number of experiments are used in the calibration process for experimental settings E and K. Note that since information gain is calculated with respect to the initial uncertainty in input parameters, the specific level of information gain, at which convergence is achieved, is irrelevant. The convergence of both the systematic bias and information gain indicates that prediction uncertainty is remedied. For the C&G model, the systematic bias converges to below 10% for the maximum stress prediction and below 50% for the texture predictions. Should this level of systematic bias be deemed acceptable for the intended purposes, the model may be considered validated, as the coverage exceeds the minimum coverage threshold. If additional reduction in systematic bias is required, however, a further physics sophistication of the model should be considered.

7.3. A Comparative Analysis: Divergence Information Criterion

In summary, according to the framework presented herein, the C&G is the preferred model for making predictions, as the G model is not validated. The lower converged discrepancy of the C&G model confirms that reduction in systematic bias can be achieved through physics sophistication of the code. However, note that due to increased complexity and a larger number of calibration parameters, the C&G model requires more experiments to reach convergence. Therefore, earlier discussion on the calibration of G and C&G models reveals a trade-off relationship between the prediction uncertainty and bias. Provided that both models are validated, preference of one model or the other can be reversed depending on the available experimental data. This section demonstrates this concept on the G and C&G models using the divergence information criterion (DIC), a statistical metric for evaluating preferable models [31].
The DIC operates on the model input parameters and their distributions to create a metric that rewards a model for predicting closer to the experiments and penalizes a model for having a larger effective number of parameters. The DIC is computed according to Eqs. (17)–(19):

$$\hat{p}_j = -2 \int \log[f^{(j)}(y|\theta^{(j)})]\pi^{(j)}(\theta^{(j)})d\theta^{(j)}$$

$$+2\log[f^{(j)}(\hat{\theta}^{(j)}(y))]$$

$$D_j(\theta^{(j)}) = -2\log[f^{(j)}(y|\theta^{(j)})] + 2\log[g^{(j)}(y)]$$

$$DIC = D_j(\hat{\theta}^{(j)}) + 2\hat{p}_j.$$  

In Eq. (17), $\hat{p}_j$ is the proposed number of effective model parameters. $f^{(j)}(y|\theta^{(j)})$ represents the probability density function for the experimental data $y$ for a given $\theta$ for the $j$th model, $\pi^{(j)}$represents the prior distribution of emulator parameters, and $\hat{\theta}^{(j)}(y)$ estimates $\theta^{(j)}$ based upon the experimental data $y$ [32]. In Eq. (18), $2\log[g^{(j)}(y)]$ is a standardizing term dependent on the observed data. In Eq. (19), $\hat{p}_j$ represents $\hat{p}_j$ when $\hat{\theta}^{(j)}(y)$ is equal to the posterior mean $\hat{\theta}^{(j)}$. When comparing two models, a smaller value for DIC indicates a preferable model. As the DIC is used for comparison purposes only, the actual DIC values are only relevant relative to another model.

Figure 13 plots the DIC values computed for G and C&G models for an increasing number of experiments. Until the fourth experiment, the less sophisticated G model has a lower DIC value and, thus, the prediction bias outweighs the uncertainty in the parameters. If more than four experiments are available, however, then the C&G model yields a smaller DIC value than that of the G model indicating that the lower prediction bias of the C&G model outweighs the penalty in higher prediction uncertainty due to a larger number of parameters.

![DIC comparison of the G and C&G models with varying levels of experimental data.](image)

**8. Conclusions**

It is a routine practice that numerical models are being validated against experiments [33, 34]. In this article, we present a framework to guide the allocation of resources for the validation of numerical models. Improvement in the predictive capabilities of a numerical model can be achieved through the reduction of prediction uncertainty and bias. The prediction uncertainty can be reduced through calibration of model parameters against experimental data. An increase in the number of experiments used in the model calibration results in a decrease in prediction uncertainty, indicated by the convergence of the information gain to a stable level throughout the domain resulting in diminishing returns from additional experiments. In this case, improvements to predictive capabilities are only possible by reducing prediction bias, which can be achieved by improving the physics and/or engineering principles of the given model.

The proposed framework is demonstrated on a nontrivial application of the VPSC code to predict stress and texture behavior of 5182 aluminum alloy. The availability of two versions of the VPSC code, G and C&G models, presents a unique opportunity to demonstrate the fundamental concepts behind the proposed resource allocation framework. In this example, for the G model, the systematic bias of the texture predictions converges while the systematic bias of the stress predictions fails to converge as more experimental data is utilized in the calibration process. The analysis of the information gain shows that parameter uncertainty cannot be further reduced. As the prediction bias for stress fails to converge and the prediction bias for texture converges to an unacceptably high value, the C&G model, with an improved constitutive law, is employed. The convergence of the systematic bias and information gain of the C&G model is observed as the number of experiments available for model calibration increases. The prediction bias of the more sophisticated C&G model converges to a smaller value than that of the G model, but requires a higher number of experiments for convergence, demonstrating the trade-off between reducing prediction bias and uncertainty in model validation. The logical thought process of the proposed framework can provide a science-based, quantifiable, and defendable rationale for allocating resources between code development and experimentation to reduce both uncertainty and bias in the predictions of complex numerical models.

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